## **IR-VUV Dielectric Function of Al<sub>1-x</sub>In<sub>x</sub>N determined by Spectroscopic Ellipsometry G6.13** A. Kasic<sup>#,1,2</sup>, M. Schubert<sup>1,2</sup>, B. Rheinländer<sup>1</sup>, J. Off<sup>3</sup>, F. Scholz<sup>3</sup>, C. M. Herzinger<sup>4</sup> <sup>1</sup>Universität Leipzig, Fakultät für Physik und Geowissenschaften, Abteilung Halbleiterphysik, 04103 Leipzig, Germany, <sup>2</sup>University of Nebraska, CMOMR, Lincoln 68588, NE, U.S.A., <sup>3</sup>Universität Stuttgart, 4. Physikalisches Institut, 70569 Stuttgart, Germany, <sup>4</sup>J.A. Woollam Co., Inc., Lincoln 68508, NE, U.S.A. #E-mail: pge95ipi@studserv.uni-leipzig.de Ellipsometry: ~ 300 ... 3000 cm<sup>-1</sup> Introduction Samples Results surface We study the infrared to deep-ultra-violet optical response of MOVPE-MOCVD. T = $750^{\circ}$ C 1), 2): We observe a one-mode 118 nm Al<sub>0.844</sub>In<sub>0.156</sub>N / 488 nm GaN / 76 nm AlN / Al<sub>2</sub>O<sub>3</sub> grown hexagonal Al<sub>1</sub>, In N films for compositions $0.110 \le x \le 0.212$ E (TO) precursors: TMAl, TMIn, NH2 behavior of the $E_1(TO)$ phonon in using Spectroscopic Ellipsometry. Al, In, N has prospects for University Stuttgart (Germany) contrast to theoretical predictions E (TO) AlInN application as confinement layer lattice matched to GaN and as active (TO) AIN using a MREI approach [1]. material for LED's and LD's operating in the NIR-VUV spectral range. AIN-like [1] The observed $Al_{1,v}In_vN E_1(TO)$ mode з - InN-like [1] shift $\Delta \omega$ with respect to unstrained a-Al<sub>1-x</sub>In<sub>x</sub>N nom, undoped 05 2 mm a-Al, In, 120 nm AlN is induced by alloy composition 04 06 08 10 0.2 x and film in-plane strain $\mathbf{e}_{x}$ : Summarv E.(TO) AlIn! a GaN buffe $\Delta w = \Delta w(x) + \Delta w(e_x) = a_0 x + a_1 x(1-x) + be_x$ 05 mm nom. undoped a•GaN buffe 1 15 mm a-Al<sub>1-x</sub>In<sub>x</sub>N 200 nm The ~120 nm thick Al<sub>1-x</sub>In<sub>x</sub>N films ① Observation of one-mode behavior of the ir-active $E_1(TO)$ phonon a-AlN buffer 80 nm 0.128 are grown pseudomorphically on in contrast to theoretical predictions [1] (0001) a-Al<sub>2</sub>O<sub>3</sub> 0001) a-Al-O (0001) a-ALO slightly strained GaN buffer layers. ① Separation of alloving and in-plane strain effects on the observed ext E (TO) AIN The ~200 nm thick Al, In N film is $E_1(TO)$ mode shift 0.12 0.15 0.18 0.21 grown fully relaxed directly on $0.117 \le x \le 0.212$ x = 0.128 $0.110 \le x \le 0.179$ 600 800 1000 ② Detection of the A<sub>1</sub>(LO) phonon mode Al<sub>2</sub>O<sub>2</sub> \_ω [cm<sup>-1</sup>] $\omega$ [cm<sup>-1</sup>] 3 Measurement of fundamental band gap energy vs. alloy composition x $a_{exp}^{AllnN} - a_{rel}^{AllnN}$ $a_{exp}^{GaN} - a_{rel}^{GaN}$ **High-resolution XRD** 0.55 µm Al<sub>0.890</sub>In<sub>0.110</sub>N / 1.5 µm GaN / Al<sub>2</sub>O<sub>3</sub> 3): The ir-ellipsometry data reveal the $A_1(LO)$ phonon mode frequency at ④ Measurement of NIR-VUV n and k spectra 850.7±0.3 cm<sup>-1</sup> for an unstrained Al<sub>0.890</sub>In<sub>0.110</sub>N film $E_1(TO)$ tensile strained $\Phi_{.} = 70$ epilaver 628.2±0.7 cm<sup>-1</sup> 8.0x10 exp (IRSE) a-Al, In N Spectroscopic Ellipsometry • [4] • [5] • [6] 4), 5): The Al<sub>1-x</sub>In<sub>x</sub>N fundamental 6.0x10 $A_1(LO)$ band gap energies as well as the 4.0x10 tensile index of refraction and the 850.7±0.3 cm<sup>-1</sup> absorption index below and above 2.0x1 the band gap were determined for exp. (IRSE - calc ••••[1] • [2], [3] different alloy compositions. room temperature 0.8 · multiple angles of incidence 0.6 -2 Ov1 rotating-compensator. x Fourier-transform based mid-infrared spectroscopic **IR-VUV Dielectric Function** ellinsometer 12 16 18 In content [%] $\omega [cm^{-1}]$ VASE®, IR-VASETM, VUV-VASETM, J.A. Woollam Co., Inc., Lincoln, NE Band gap Assuming Vegard's law, the composition x follows from the measured c $E_{\rm c}({\rm TO})$ 4.52 eV and a-lattice constants. a-Al0820In0180N is expected to be lattice matched Ellipsometry: ~1 eV ... 8.5 eV 637.7 cmto strain-free a-GaN **Data Analysis** 346 nm Al<sub>0.827</sub>In<sub>0.173</sub>N / 882 nm GaN / Al<sub>2</sub>O<sub>3</sub> 196 nm Al<sub>0.872</sub>In<sub>0.128</sub>N / Al<sub>2</sub>O<sub>3</sub> Comparing measured and calculated data through regression data **Raman Scattering** analysis: lineshape fit parameterized film dielectric function and film thickness $\lambda_{ex} = 514.5 \text{ nm}$ $P_{ux} = 100 \text{ mW}$ [a.u.] mid-IR parameters intensity TO frequency Л LO frequency mode broadening phonon free-carrier $j = ||, \perp c$ -axis 118 nm a-Al<sub>0.813</sub>In<sub>0.187</sub>N high-frequency dielec. function limit properties properties 561 nm a GaN (unstrained) 0.08 0.16 2 4 6 NIR-VUV 0.00 84 nm a AlN screened plasma 196 nm Al<sub>0.872</sub>In<sub>0.128</sub>N Energy [eV] frequency a Al<sub>2</sub>O<sub>2</sub> Al<sub>2</sub>O<sub>2</sub> plasma broadening kth osc. amplitude 600 H. Grille, Ch. Schnither, F. Becheneb, Phys. Rev. B 40, 6991 (2000) W. Dorspher, et al., Phys. Rev. B 50, 705 (1990) W. Dorspher, et al., Phys. Rev. B 50, 705 (1990) W. Dorspher, et al., Phys. Rev. B 50, 705 (1990) W. Dorspher, Phys. Rev. B 50, 705 (1990), and phys. Rev. B 50, 705 (1990) W. Tay, D. Schon, Appl. Phys. Lett. 6, 3460 (1995), aborption messurements K. S. K. M. A. Scher, P. Kung, M. Rarging, K. Y. Lim, Appl. Phys. Lett. 71, 800 (1997), aborption $e^{j}(w) = e^{-.j} + \sum_{2}$ kth osc. frequency Band gap ω [cm<sup>-1</sup> C. exp[...]k<sup>th</sup> osc. broadenin oscillator E kth osc. phase facto Energy [eV] Energy [eV] Micro-Raman scattering measurements do not reveal any Al, In,N band-gap narameters phonon modes for layer thicknesses $d \le 0.2 \text{ mm}$ .

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